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# Polymers from Renewable Resources. X. Semi-Interpenetrating Polymer Networks Based on Castor Oil Polyurethane and Cardanol-Furfural Resin: Scanning Electron Microscopy and XRD Studies 

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# POLYMERS FROM RENEWABLE RESOURCES. X. SEMIINTERPENETRATING POLYMER NETWORKS BASED ON CASTOR OIL POLYURETHANE AND CARDANOLFURFURAL RESIN: SCANNING ELECTRON MICROSCOPY AND XRD STUDIES 

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## ABSTRACT

A number of semi-interpenetrating polymer networks (Semi-IPNs) were synthesized by reacting the polyurethane prepared from castor oil and diisocyanates and the resin prepared from cardanol and furfural. The physico-chemical properties of the semi-IPNs have been investigated. The scanning electron microscopy of some of the semi-IPNs have been studied and the morphology has been examined. The samples were subjected to wide angle $X$-ray diffraction analysis. The degree of crystallinity ( $X_{C r}$ ) were computed on the basis of crystal defect concept developed by Roland and Vonk.

## INTRODUCTION

The research work involving interpenetrating polymer networks (IPNs) using naturally occuring triglyceride oils which was initiated by Sperling and Coworkers (1-5) at Leigh University has taken momentum in recent years. Over the years a large number of papers have been published in this area using biomonomers as the base materials. These oils are generally synthesized by the nature with multiple chemical functionality. The oils which have attracted attention for preparation of IPNs and semi-IPNs are castor, vernonia and lesquerella palmeri etc. Out of these oils, castor oil has received attention in recent years because of its availability in large commercial scale. Orissa is one of the states in India crowned with various forest products of oil bearing wild plants such as castor, tung, vernonia, linseed, crambe, lunaria, cashewnut, hipatage benghalensis, wrightia tinctoria, w.tomentosa and apocynaceace. These plants are abundantly grown in the forest of orissa and the oils of some
of these plants contain trigleceride of recenoleic acid. Thi ; laboratory has taken up a program for the synthesis and characterization of TPNs and semi-IPNs from the oils available in the forests of Orissa. In the first step we have reported the characterization of some IPNs from castor oil and cashewnut shell liquid, a by-product of cashew industries (6-7). This communication presents the scanning eleztron microscopy and XRD study of some semi-IPNs prepared from castor oil polyurethane and cardanol-furfural based resins.

## EXPERIMENTAL

Synthesis of Polyurethane (Polymer-I) : Castor oil (9.32gm, 0.01 mol) was allowed to react with toluene 2 , 4 -diisocyanate (TDI) ( $4.18 \mathrm{gm}, 0.024$ mole) to maintain the $\mathrm{NCO} / \mathrm{OH}$ ratio at 1.6. The reaction was carried out at $45^{\circ} \mathrm{C}$ with continuous stirring for 2 hours. The prepolymer was isolated as a viscous liquid. Other polyurethanes with different $\mathrm{NCO} / \mathrm{OH}$ ratios were prepaired following the same procedure and also differing with diisocyanate.

Synthesis of Resin (Polymer-II) : A mixture of cardanol (4 m.mole) and furfural ( $40 \mathrm{~m} . \mathrm{mole}$ ) was refluxed at $110^{\circ} \mathrm{C}$ for abcut 8-10 hr in presence of NaOH (4N) with periodical shaking. At the end of reaction, the contents extracted with ice cold water, filtered and washed repeatedly with hot water to remove the unreacted materials. The product was dried in vacuum.

Synthesis of semi-IPNs: These components polymer- I and II were taken in various ratios (shown in Table-l) in benzene. Then 5cc of $1 \%$ EGDM (ethylene glycol dimethacrylate) was added to it and stirred well on electric stirrer. The string was continued till the reactions are intimately mixed. The entire process was carried out at ambient. temperature. These two polymers continue simultaneously polymerization forming semi-interpenetrating polymer networks. The physico-chemical properties of the semi-TPNs are represented in Table-l \& 2.

Scanning electron micrographs were obtained from a phillips EM-400 equipped with scanning (transmission) electron mi:roscope system PW6585. The film was cryogenically fractured in liquid nitrogen and was mounted vertically on a SEM stub by silver adhesive paste. The specimen was coated wi th gold using an EPS/carl zeiss vacuum system at about 10 to r .

Wide anglle $X$-ray diffraction data were obtained using $\mathrm{CuK}_{\alpha}$ radiation which was monochromatized by curved quartz crystal after transmission through the specimen. Intensity distribution was recorded using symmetrical reflection teshnique in a phillips diffractomer (PWll40). $\mathrm{CuK}_{\alpha}$ (1.54A)

TABLE-1
Feed composition and physicochemical data of semi-IPNs

| Sl. | Sample <br> Code | Resin <br> Code | COPU <br> (NCO/OH) | COPU/ <br> Resin | $\left.\begin{array}{l}\text { Decompo- } \\ \text { sition } \\ \text { tempera- } \\ \text { ture }\end{array}{ }^{\circ} \mathrm{C}\right)$ |
| :--- | :--- | :--- | :--- | :--- | :--- | :--- |

$(1-4) \rightarrow$ COPU- Castor oil polyurethane from DPMDI (diphenyl methane diisocyanate), CARD - Cardanol, FU - Furfural, $(5-8) \rightarrow$ COPU $\rightarrow$ Castor oil polyurethane from TDI (toluene 2,4-diisocyanate).

TABLE-2
Percentage of weight loss on treatment with different chemical reagents

| Chemical Reagents | $\begin{aligned} & \hline \text { Semi- } \\ & \text { IPN-1 } \end{aligned}$ | $\begin{aligned} & \text { Semi- } \\ & \text { IPN-2 } \end{aligned}$ | $\begin{aligned} & \text { Semi- } \\ & \text { IPN-3 } \end{aligned}$ | Semi- IPN-4 | $\begin{aligned} & \text { Semi- } \\ & \text { IPN-5 } \end{aligned}$ | $\begin{aligned} & \text { Semi- } \\ & \text { IPN-6 } \end{aligned}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| $25 \% \mathrm{H}_{2} \mathrm{SO}_{4}$ | 1.0 | 1.0 | 1.1 | 1.0 | 1.1 | 1.1 |
| $25 \% \mathrm{HCl}$ | 1.7 | . 0.3 | 0.7 | 1.5 | 1.5 | 1.3 |
| 40\% NaCl | 0.7 | 0.2 | 0.1 | 0.8 | 0.9 | 2.4 |
| $5 \% \mathrm{NaOH}$ | 1.8 | 9.2 | 12.0 | 10.3 | 12.4 | 1.7 |
| $5 \% \mathrm{HNO}_{3}$ | 1.8 | 0.5 | 0.6 | 1.6 | 1.6 | 2.3 |
| $10 \% \mathrm{NH}_{4} \mathrm{OH}$ | 0 | 0 | 0 | 0 | 0.8 | 0 |
| Distilled water | 0.7 | 0.7 | 0.6 | 0.7 | 0.5 | 0.4 |

radiation at 30 KV , l0MA with balanced filter of $\mathrm{Ni}-\mathrm{Al}$ was used. Intensity measurement were taken using G.M counter employing fixed count step scanning Platinium powder was used as instrument standard.

## XRD STUDIES

X-ray diffraction pattern of many polymers consisting both sharp and diffused ring reveals their two phase morphology. The high mechanical strength of these polymers are attributed to the presence of crystalline domains in the composite two phase microstructure. The determination of crystalline content of a given samples accurately, has been the subject matter of great interest for the last few decades in order to get a first hand knowledge about the performance of a polymer materials. Hermons (8) and Karst and Flaschner (9) developed quantitative methods for determination of the degree of crystallin (Xcr). These two methods define crystalline domains as those which contribute maxima in its $X$ ray diffraction curve and arrived at a crystalline proportion. By comparison of the intensity of maxima considered to be contributed by the crystalline part of the substance and the intensity of the background scattering supposed to be due to amorphous regions. Ruland (10) developed a method for the determination of. ( $X_{c r}$ ) by introducing crystal defect concept which was subsequently simplified and computered by Vonk(11). Hindleh (12) developed a poak resolution technique to find out $X_{c r}$ values. The present study is basing on the determination Xcr of semi-IPNs prepared from castor oil based polyurethanes and cardanol-furfural based resins by Ruland-Vonk crystal defect method. Crystalline size can be determined by Scherrer's formula (13).

Ruland-Vonk method : Ruland (10) proposed the following equation based on the crystal defect concept.

$$
\begin{equation*}
x_{c r}=\frac{1}{R} \times k=\frac{\int_{S_{O}}^{S_{p}} I_{C r}(s) s^{2} d s}{\int_{S_{O}}^{S_{p}} I(s) s^{2} d s} \int_{S_{0}}^{S_{0} \bar{f}^{2}\left(s^{2}\right) d s} \int_{S_{p}}^{S_{p} f^{2}\left(s^{2}\right) D . d s} \tag{1}
\end{equation*}
$$

The term ' $\frac{1}{R}$ ' refers to conventional ' $X_{c r}$ ' and ' $K$ ' is a crystal defect constant, 'Icr' is the part of the intensity in the crystalline lattice vector $S=2$ sino/f(for Bragg's angle (20) and wave length $\lambda$ ). 'I' is the total coherent scattering at point's', $\mathbf{f}^{2}$ is the mean scattering factor

$$
f^{2}=\frac{\left\langle N_{i} f_{i}^{2}\right.}{\sum N_{i}}
$$

where ' $\mathrm{fi}_{\mathrm{i}}$ ' is the atomic scattering factor of an atom of the type 'i' and ' Ni ' is the number of atoms of the type 'i', 'D' is the disorder function which taken into account the loss of intensity concentrated at the reciprocal lattice point due to deviations of the atoms from their ideal positions. These deviation may be due to thermal vibration of the first or second kind

$$
\begin{equation*}
D=2 \exp \left(-a s^{2}\right) / 1+\exp \left(-a s^{2}\right) \tag{2}
\end{equation*}
$$

where 'a' is the overal distortion factor due to temperature and lattice distortions. One of the problems in the evaluation of $\mathrm{X}_{\mathrm{cr}}$ by this method is the large amount of calculation involved particularly in selecting the interval until $X_{\text {cr }}$ remains constant while varying $K$ (and hence $D$ ). Again knowledge of the exact chemical structure of the compound and unit cell parameters are necessary. Vonk simplified and computerised the Ruland's method. According to this method a plot of Rsp versus $S_{p}$ will oscillate about a straight line defined by

$$
\begin{equation*}
R_{s p}=1 / X_{c r}+\left(K / 2 X_{c r}\right) s p p^{2} \tag{3}
\end{equation*}
$$

where 'sp' is the upper integration limit. Xer is estimated from the intercept and ' K ' from the slope. Thus we could generate a set of different 'Rsp' values and their corresponding 'Sp' values for getting a plot of Rsp Vs sp. In this computation we have taken the help of the LOTUS 123 software in a PCAT. Table -3 shows the representative data and calculation results of semi-IPN-5. A series of Rsp values for different upper limit (sp) have been found out. A plot of Rsp Vs sp was obtained from various Rsp values and their corresponding s? values. As expected, it will give an oscillating curve through which a straight line could be drawn. The intercept and the slope could be determined by least square method using regression analysis in the LOTUS 123 software. The intercept as described in equation (1) gave the values of $\left(1 / X_{C r}\right)$ and the reciprocal of this value gave Xer.

## SCANNITGG ELECTRON MICROSCOPY (SEM) OF SEMI-IPNs

The morphology of the semi-IPNs derived from castor oil hased polyurethanes and cardanol-furfural based resins have been studied by scanning electron microscope (SEM) method. Fig.l(a) \& (b) shows the SEM micrographs of the semi-IPN-2 prepared from castor oil-DPMDI polyurethane ( $\mathrm{NCO} / \mathrm{OH}$ - 1.8 ) and cardanol-furfural resin (35:65) with 2500 to 5500 magnification respectively. A heterogeneous surface was observed. Fig.2(a) (b) shows the SEM micrographs of the semi-IPN-3 prepared from castor oil-DPMDI polyurethane (NCO/OH - 1.8) and cardanol-furfural (45:55) with 1500 to 2500 magnification respectively. By comparing both the figures, it is observed that the heterogeneity gradually
TABLE - 3

| 20 | I | $S_{p}$ | $I_{s}{ }_{p}^{2}$ | s? | $\mathbf{A}_{\text {IW }}$ | $\mathbf{A}_{\mathrm{IOB}}$ | $\begin{gathered} A_{I A B} \\ \left(A_{I W}^{-A_{I U B}}\right) \end{gathered}$ | $\mathbf{A I N}^{\text {I }}$ | $\boldsymbol{A}_{\text {IAB }}$ | $\mathbf{R}_{\mathbf{p}}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| 10 | 0.3 | 0.1131 | 0.7996 | 0.0127 | 0.0099 | 0.0083 | 0.0016 | 0.0095 | 0.0016 | 6.1888 |
| 11 | 0.3 | 0.1243 | 0.9671 | 0.0154 | 0.0119 | 0.0114 | 0.0005 | 0.0218 | 0.0021 | 10.3466 |
| 12 | 0.3 | 0.1356 | 1.1502 | 0.0184 | 0.0155 | 0.0144 | 0.0011 | 0.0374 | 0.0032 | 11.6447 |
| 13 | 0.35 | 0.1469 | 1.6189 | 0.0215 | 0.0231 | 0.0175 | 0.0056 | 0.0606 | 0.0088 | 6.8502 |
| 14 | 0.4 | 0.1581 | 2.5017 | 0.0250 | 0.0321 | 0.0205 | 0.0116 | 0.0928 | 0.0204 | 4.5365 |
| 15 | 0.45 | 0.1694 | 3.2285 | 0.0286 | 0.0432 | 0.0236 | 0.0196 | 0.1361 | 0.0401 | 3.3909 |
| 16 | 0.6 | 0.1806 | 4.4860 | 0.0326 | 0.0586 | 0.0266 | 0.0320 | 0.1947 | 0.0721 | 2.6987 |
| 17 | 0.65 | 0.1918 | 5.9801 | 0.0368 | 0.0796 | 0.0296 | 0.0499 | 0.2744 | 0.1221 | 2.2461 |
| 18 | 0.85 | 0.2030 | 8.2442 | 0.0412 | 0.1037 | 0.0326 | 0.0711 | 0.3781 | 0.1933 | 1.9561 |
| 19 | 1.45 | 0.2142 | 9.6241 | 0.0458 | 0.1178 | 0.0355 | 0.0822 | 0.4960 | 0.2756 | 1.7998 |
| 20 | 1.3 | 0.2253 | 10.193 | 0.0507 | 0.1302 | 0.0385 | 0.0917 | 0.6263 | 0.3673 | 1.7050 |
| 21 | 1.3 | 0.2365 | 9.7864 | 0.0559 | 0.1382 | 0.0414 | 0.0967 | 0.7646 | 0.4641 | 1.6474 |
| 22 | 1.21 | 0.2476 | 9.2654 | 0.0613 | 0.1331 | 0.0444 | 0.0887 | 0.8978 | 0.5529 | 1.6237 |
| 23 | 0.9 | 0.2587 | 8.7166 | 0.0669 | 0.1255 | 0.0473 | 0.0782 | 1.0233 | 0.6311 | 1.6214 |
| 24 | 0.85 | 0.2698 | 8.3220 | 0.0728 | 0.1150 | 0.0501 | 0.0648 | 1.1383 | 0.6959 | 1.6356 |
| 25 | 0.7 | 0.2809 | 7.8636 | 0.0789 | 0.1015 | 0.0530 | 0.0485 | 1.2399 | 0.7444 | 1.6655 |
| 26 | 0.6 | 0.2919 | 6.3237 | 0.0852 | 0.0912 | 0.0558 | 0.0353 | 1.3311 | 0.7798 | 1.7070 |
| 27 | 0.5 | 0.3029 | 6.8321 | 0.0917 | 0.0848 | 0.0587 | 0.0261 | 1.4160 | 0.8059 | 1.7569 |
| 28 | 0.45 | 0.3139 | 6.3937 | 0.0985 | 0.0840 | 0.0615 | 0.0225 | 1.5000 | 0.8285 | 1.8105 |
| 29 | 0.4 | 0.3249 | 6.9198 | 0.1055 | 0.0897 | 0.0642 | 0.0254 | 1.5897 | $0.8539$ | $1.8616$ |
| 30 | 0.35 | 0.3359 | 5.8626 | 0.1128 | 0.0955 | 0.0670 | 0.0285 | 1.6853 | 0.8824 | 1.9098 | and

$\begin{aligned} & \text { varying sp values e.g. area under the curve upto AF), AIAB }=\text { summation } \\ & \text { of individual areas above the base line (area between fixed initial so }\end{aligned}$
and varying $s_{p}$ above the base line such as area above the line AQ).
$\begin{aligned} & \text { corresponding to 'sp', ATW }=\text { individual whole area (for varying lower } \\ & \text { and upper limits) e.g area of trapezium BCEF, ATUB }=\text { individual area }\end{aligned}$
under the base line (for varying lower and upper inmit e g area of
trapezium PQEC), A $A$ individual area above the base line
Cr
.
KEY FORDS :-
-
$x=0.677229$
$\longrightarrow \longrightarrow$ -
apezium PQEC), $A_{I A B}=$ individual area above the base line,
IAB $=$ individual area above the base line,

$$
\begin{aligned}
& \text { KEY WORDS }
\end{aligned}
$$



Fig. 1 (a) SEM Micrograph of Semi-IPN-2 with magnification 2500
(b) SEM Micrograph of Semi-TPN-2 with magnification 5500


Fig. 2 (a) SEM Micrograph of Seni-IPN-3 with magnification 1500 (b) SEM Micrograph of Semi-IPN-3 with magnification 2500


Fig. 3 (a) SEM micrograph of Semi-IPN-6 with magnification 1500.
(b) SEM micrograph of Semi-IPN-6 with magnification 3000.
decreases and the morphology gradually changes from discontinuous to continuous with increaging the resin content from 55\% to 65\%. Similarly varying the COPU-resin content constant when NCO/OH ratio of the polyurethane changes from 1.6 to 2 again the heterogenity becomes more prominent. This indicates that with increasing the resin content, the system becomes homogeneous as is evident from the nature of crystallinity in case of XRD study. Similarly observations have also been observed in case of the semi-IPN-6 prepared from castor oil-TDI polyurethane ( $\mathrm{NCO} / \mathrm{OH}-1.8$ ) and cardanolfurfural resin (35:65) with 1500 to 3000 magnification which is shown in the fig. $3(a)$ ( $b$ ).

It is observed that the semi-IPN-4prepared from castor oil-DPMDI (NCO/OH-2) with the resin copolymer (45:55) has the highest degree of crystallinity ( $\mathrm{X}_{\mathrm{Cr}}$ ) i.e. 0.73772. The degree of crystallinity value decreases to 0.567738 as the $\mathrm{NCO} / \mathrm{OH}$ ratio of the polyurethane decreases from 2 to 1.6 . Hence in this case the crystallinity depends on the $\mathrm{NCO} / \mathrm{OH}$ ratio of the polyurethane rather than copu resin ratio. Similar observation has also been observed in case of the polyurethane prepared from TDI. The order of crystallinity of different samples under investigation are furnished in Table- 4 . From the observation, it can be ascertained that

## TABLE-4 <br> Degree of Crystallinity ( $\mathrm{X}_{\mathbf{c r}}$ ) of Polymer Samples

| S1.No. | Sample No. | X $_{\text {Cr }}$ |
| :--- | :--- | :---: |
| 1. |  |  |
| 2. | Semi-IPN-1 | 0.567738 |
| 3. | Semi-IPN-3 | 0.639362 |
| 4. | Semi-IPN-4 | 0.737720 |
| 5. | Semi-IPN-5 | 0.677229 |
| 6. | Semi-IPN-7 | 0.681324 |
|  | Semi-IPN-8 | 0.849149 |



Fig. 4 Intensity distribution plot of semi-IPN-5
most probably the resins one of comparatible with the polyurethanes and a two phase system exists.

## RESULT AND DISCUSSION

In the present investigation Ruland and Vonk method has been employed for calculating the degree of crystallinity ( $\mathrm{X}_{\mathrm{cr}}$ ) values. The detailed method of computing Xcr for the semi-TPN-5 is furnished in Table-3. In fig.(4) the XRD intensity distribution plot for the semi-IPN- 5 which plots Is ${ }^{2} \mathrm{Vs} \mathrm{S}$ is furnished.


Fig. 5 : Intensity (I) Vs Bragg's angle (20), (a) semi-IPN-8 $(\infty)$, (b) semi-IPN-7 ( $\Delta$ ), (c) semi-IFN-5 (口), (d) semi-IPN-4 (十), (e) semi-IPN-1 (/).

This plot has been obtained from the plot of intensity (I) Vs Bragg's angle (20) is shown in fig. (5). The plot indicate the presence of two sets of reflection planes corresponding to the (2Q) values of $14^{\circ}$ and $28^{\circ}$. Since our aim is to draw a plot of Rsp Vs sp different upper limit (sp) values had to be chosen. This is more clearly observed in fig.(4). Let'So' is the lower limit and 'sp' is the first upper limit. The area bounded by the curve upto ' $S_{p}$ ' can be found out by calculating the area of the trapezium $A B C D$. The area below the background line between so to sp is the area of the trapezium APCD. The area above the base line which is predominantly due to the crystaline domain obtained by sunstracting the area of the trapezium $A B C D$.

The ratio of the areas of triangle $A B P$ and trapezium $A B C D$ is $1 / R_{s p}$ and hence the reciprocal of this value is $R_{s p}$ and this is restricted to the limits so to $s_{p}^{1}$. Now a new upper limit sp whose value is, higher than ' $S_{p}$ ' js taken. The area below the curve So to Sp is found out as follows. First the area of the trapezium $B C E F$ and $P Q E C$ are calculated between limits Sp to $S_{p}^{\prime}$ as before. These areas added to the previous area of trapezium $A B C D$ and $A P C D$ respectively to find out the total area under the curve and below the base line between the limits $S_{0}$ to $S_{p}$. The total area above the background line betwsen so to $s^{\prime} p$ is obtained by substracting the total area area below the line from the total area under the curve
between the limits So to $s_{p}^{\prime}$. In this manner the higher limits that is $S p$ values were increased and the total areas between So to Sp is obtained by summing up technique. The $R_{\text {sp }}$ value at each limit could be calculated by taking the ratio of the areas under the whole curve and above the base line between these limits.

$$
R_{s p}=\frac{\int_{s_{0}}^{s_{p}} I(s) s^{2} d s}{\int_{s_{0}}^{s_{p}} I_{c r}(s) s^{2} d s}
$$

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